



## Research article

# Experimental and theoretical study of improved mesoporous titanium dioxide perovskite solar cell: The impact of modification with graphene oxide

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## ABSTRACT

The present study serves experimental and theoretical analyses in developing a hybrid advanced structure as a photolysis, which is based on electrospun Graphene Oxide-titanium dioxide (GO-TiO<sub>2</sub>) nanofibers as an electron transfer material (ETMs) functionalized for perovskite solar cell (PVSCs) with GO. The prepared ETMs were utilized for the synthesis of mixed-cation (FAPbI<sub>3</sub>)<sub>0.8</sub>(MAPbBr<sub>3</sub>)<sub>0.2</sub>. The effect of GO on TiO<sub>2</sub> and their chemical structure, electronic and morphological characteristic were investigated and discussed. The elaborated device, namely ITO/BL-TiO<sub>2</sub>/3 wt% GO-TiO<sub>2</sub>/(FAPbI<sub>3</sub>)<sub>0.8</sub>(MAPbBr<sub>3</sub>)<sub>0.2</sub>/spiro-MeTAD/Pt, displayed 20.14% disposition and conversion solar energy with fill factor (FF) of 1.176%, short circuit current density (J<sub>sc</sub>) of 20.56 mA/cm<sup>2</sup> and open circuit voltage (VOC) 0.912 V. The obtained efficiency is higher than titanium oxide (18.42%) and other prepared GO-TiO<sub>2</sub> composite nanofibers based ETMs. The developed materials and device would facilitate elaboration of advanced functional materials and devices for energy storage applications.

## 1. Introduction

Recently, perovskite solar cells (PVSCs) have attracted huge attention due to the fast-growth of 3.8%–25.2% power conversion

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efficiency (PCE) within a very short period, which now compete with the silicon photovoltaics [1]. Organic-inorganic perovskites have been consistently attracting attention of the scientists and engineers alike in the photovoltaic society in view of their extraordinary features, e.g. long diffusion length and carrier lifetimes [2], such that the maximum conversion in this type of device approached 25.5% [3]. To improve their efficiency, some various methods or strategies were followed, e.g. the carrier multiplication impact, to get the more energy of photon with the energy larger than bandgap and multijunction absorbers to get photons with smaller energy than bandgap [4]. Presently, two kinds of architectures are known and practiced in solar cells. The first is a solar cell with a layer of mesoporous (mp) electron transport contain from semiconductor metal oxide such as  $\text{TiO}_2$ . The second type works on the basis of position of electrons and holes transporting within cell structure, which is classified as of p-i-n or n-i-p junction planer structure [5,6]. Particularly, for the solar cell with mesostructured, a solution-treated methylammonium lead iodide ( $\text{CH}_3\text{NH}_3\text{PbI}_3$ ) shows diffusion length of electron ( $e^-$ ) and hole ( $h^+$ ) of 100 and 130 nm approximately. Recently, by using electron beam induced current imaging method, Edri et al., illustrated that the diffusion length of hole is longer than that of electron in titanium oxide ( $\text{TiO}_2$ ) based electron transport layers [7]. Typically,  $\text{TiO}_2$  has been employed in designing mesoporous electron transport layer (mp-ETL) [8,9]. However, the recombination of charges in mesoporous  $\text{TiO}_2$  based electron transport layer is still remains a challenge as it inhibits the transfer of electron because of the scattering of grain boundary, which limits conversion of energy [10–16]. Therefore, enhancing the transport of charge has become an acquisition interest. For the mesostructured cells, many endeavors have been made to expedite transport of charges via using incorporated  $\text{TiO}_2$  or substituting it by diverse ions such as  $\text{Al}^{3+}$ ,  $\text{Nb}^{5+}$ ; while morphology change to nanowires  $\text{TiO}_2$ , rods and fibers was also studied [17–20]. On the other side, graphene oxide (GO) has an uncommonly high mobility of electron, high lucidity at room temperature, and higher stability, which makes it an appropriate substitute to enhance the overall efficiency of conversion for energy devices such as photochemical cells, supercapacitor and batteries [21,22–28]. Thus, GO has been utilized in various layers of PVSCs, like in mesoporous  $\text{TiO}_2$  or even utilized in hole transfer materials [29–33]. Cho [34,35–40] served rGO- $\text{TiO}_2$  as a layer in PVSCs and enhanced Open Circuit Voltage (VOC) and Fill Factor (FF), resulting in high efficiency with 19.54% PCE. Wong [41] reported the use of GO- $\text{TiO}_2$  layers in PVSCs working under low temperature processes with 15.6% PCE. There are several more examples of using GO- $\text{TiO}_2$  inducing positive effects on PVSCs performance [42–46]. From above literature, it is evident that use of GO could enhance the electronic properties of electron transfer materials [29,30,47–50]. Between all precipitation methods, electrospinning has engaged more interest because of its ability to synthesize nanoparticles with high aspect and porous [51]. Moreover, electrospinning method ease the short fabrication time, nanomaterial can be regulated easy and ecofriendly, mainly considered in biomedical engineering [52,53–55].

Herein, we aim to show a performance enhancement of mesostructured PVSCs by using GO/mp- $\text{TiO}_2$  as electron transport layer (ETL) with  $(\text{FAPbI}_3)_{0.8}(\text{MAPbBr}_3)_{0.2}$  perovskite absorber used for effective PVSCs. The impact of GO on the performance of perovskite solar cell was estimated via time-resolved photoluminescence and ephemeral measurements. Furthermore, that the performance of GO/ $\text{TiO}_2$  through reducing the interfacial resistance was compared with that of mp- $\text{TiO}_2$ , as measured in terms of improvement in the efficacy of charge collection.

## 2. Experimental

### 2.1. Materials

Titanium isopropoxide, methyl amine, diethyl ether and acetic acid were supplied from Merck Co., while the polyvinylpyrrolidone

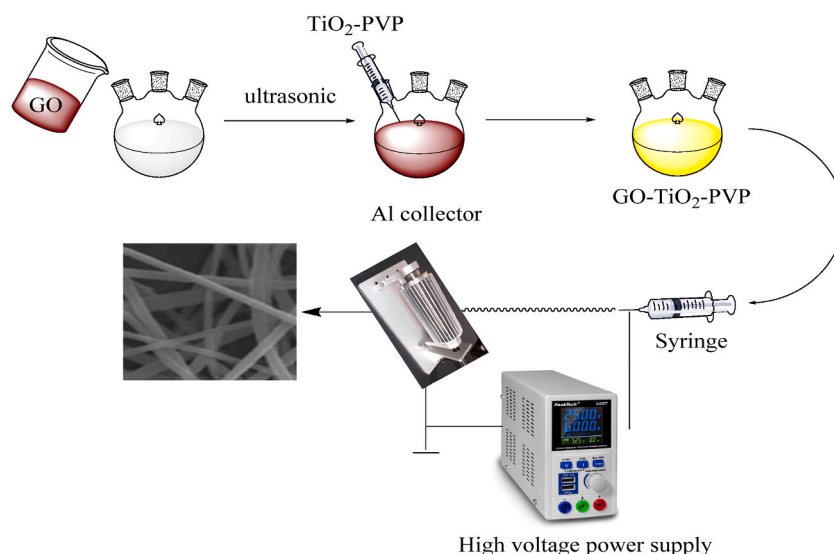


Fig. 1. Schematic of preparation of electrospun  $\text{GO-TiO}_2$  NFs.

(PVP, M.wt 16000), 2,2',7,7'-tetrakis-(N,N-di-p-methoxyphenyl-amine)-9,9', formamidine acetate, chlorobenzene, hydroiodic acid, hydrobromic acid, bis (trifluoromethane) sulfonamide lithium salt, 4-tert-butylpyridine and ethanol were purchased from Sigma Aldrich(99%). All the aforementioned chemicals were used without future purification. Fluorine tin oxide conductive glass sheets was supplied by Pilkington Co.

## 2.2. Synthesis of GO-TiO<sub>2</sub> nanofibers (NFs)

The synthesis of GO-TiO<sub>2</sub> NFs was carried out using manual electrospinning system (Fig. 1). The electrospinning system was contained from two electrodes to connect with syringe pump and rotating drum that covered by aluminum foil under 450 rpm speed. The distance between anode and cathode was 15 cm, and the feeding rate was 1 ml/h and the applied voltage between the tip needle to rotating is 9 kV. Initially, in 3 ml absolute ethanol, 5 mg of GO was dispersed by using ultrasonication process [56,57]. After that, it was mixed with 50 ml of colloidal yellow TiO<sub>2</sub>-PVP ethanolic viscous solution. Then, the mixture was transferred carefully to the syringe pump connected with positive electrode, while the negative electrode was linked with rotating drum. The obtained PVP-GO-TiO<sub>2</sub> NFs precipitated on the covered aluminum foil. Finally, it strictly removed via pointed force and calcined at 400 °C for 1 h donated as GO-TiO<sub>2</sub> NFs. The same experimental process was repeated to prepare different weight ratio of GO-TiO<sub>2</sub> designated as 3 wt% GO-TiO<sub>2</sub> and 5 wt% GO-TiO<sub>2</sub>. The TiO<sub>2</sub> and GO-TiO<sub>2</sub> were prepared in required quantity of ethanol and ethyl cellulose and utilized for spin coating. The thickness of paste was controlled via ethanol dilution and spin coating. In present experiments, the optimized conditions were coated via 4500 rpm for 30 s with volume ratio of 1:5 v/v. Finally, prepared samples calcined at 550 °C for 1 h and cooled at room temperature.

## 2.3. Device fabrication

The indium tin oxide (ITO) conductor glass with dimension 3 × 2 was ultrasound in absolute ethanol for 30 min before rinsed and dried using hot air flow. The carbon-titanium dioxide (C-TiO<sub>2</sub>) was precipitated on the ITO conductive side as follows: the solution was prepared via mixing 10 ml absolute ethanol, 1 ml titanium chloride and 0.5 sulphuric acid (98%). Then, ultrasonic it for 1h, and was coated on the ITO substrate at 3000 rpm for 2 min using spin coating process. After that, it was dried for 1h at room temperature for 30 min, and posteriorly burned at 500 °C for 1h to prompt the crystallization. The layer of 0.01%wt. GO/TiO<sub>2</sub> was precipitated on the C-TiO<sub>2</sub> layer as follow: the precursor solution of 0.04%wt. GO/TiO<sub>2</sub> was prepared via mixing TiO<sub>2</sub>, GO and absolute ethanol (1:4 wt/wt. ratio). The solution was ultrasound for 30 min and spin-coated (0.1 ml) on the layer of TiO<sub>2</sub> at 300 rpm for 1 min, and then heated it at 250 °C for 1h. To obtain perovskite (FAPbI<sub>3</sub>)<sub>0.8</sub>(MAPbBr<sub>3</sub>)<sub>0.2</sub> layer, a method with one step was done under N<sub>2</sub> conditions. In 100 ml of DMSO/DMF solvents, Formamidinium iodide (FAI) prepared by condensing 30 ml HI with 14g of C<sub>3</sub>H<sub>8</sub>N<sub>2</sub>O<sub>2</sub>, methylammonium lead bromide prepared by condensing 5 ml HBr with 40 wt% methyl amide, lead iodide and lead bromide were mixed together under ice condition. The mix solution was spin-coated on 0.01 %wt. GO/TiO<sub>2</sub> layer at 5000 rpm through 60 s and followed via drying at 80 °C for 1 h. For (spiro-OMeTAD) layer, 0.16 g of 2,2',7,7'-tetrakis-(N,N-di-p-methoxyphenyl-amine)-9,9'-spirobifluorene in 2 ml chlorobenzene, 76 ml of bis (trifluoromethane) sulfonamide lithium salt in 30 ml 4-tert-butylpyridine was mixed with strong stir for 2h. The (spiro-OMeTAD) solution was spin-coated at 4000 rpm for 1 min and leave it for 3h to enhance the conductivity. Finally, the prepared device was carried to chamber of vacuum evaporation (1 × 10<sup>-6</sup> torr) for precipitation of Pt contact that prepared by photolysis

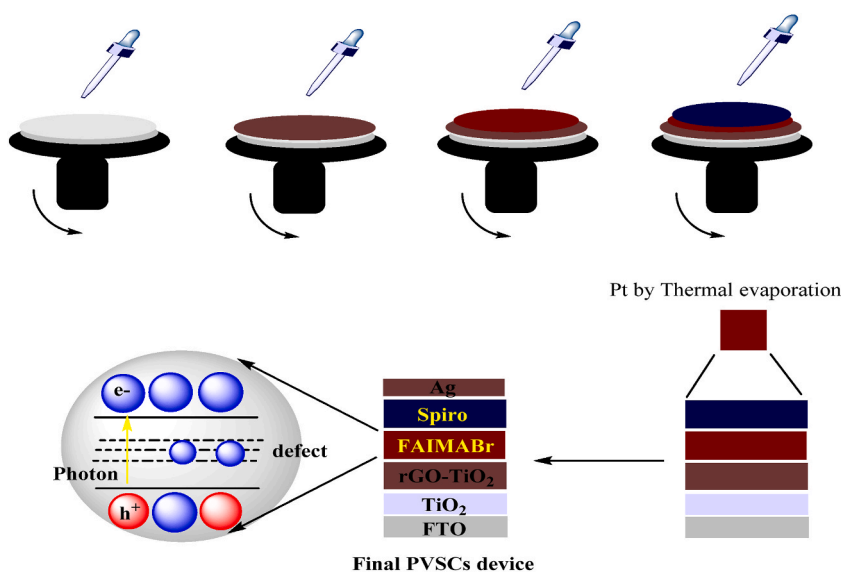


Fig. 2. Steps taken in the fabrication of PVSCs.

method [58–60,61–67] previously. The temperature of chamber was maintained unchanged by putting the source material in contact with a hot surface, that was resistively heated by passing a current through it. The fabricated process was summarized in Fig. 2.

#### 2.4. Theoretical details

The PVSC with (n-i-p) configuration (ITO/TiO<sub>2</sub>/GO-TiO<sub>2</sub>/(FAPbI<sub>3</sub>)<sub>0.8</sub>(MAPbBr<sub>3</sub>)<sub>0.2</sub>/spiro-OMeTAD/Pt) utilized for numerical emulation is demonstrated in Fig. 3a. A model explanation of a cross section with each layer for (ITO/TiO<sub>2</sub>/GO-TiO<sub>2</sub>/(FAPbI<sub>3</sub>)<sub>0.8</sub>(MAPbBr<sub>3</sub>)<sub>0.2</sub>/spiro-OMeTAD/Pt) is displayed in Fig. 3b. To comprehend the behavior of (ITO/TiO<sub>2</sub>/GO-TiO<sub>2</sub>/(FAPbI<sub>3</sub>)<sub>0.8</sub>(MAPbBr<sub>3</sub>)<sub>0.2</sub>/spiro-OMeTAD/Pt) solar cell, a theoretical study was carried out using SCAPS-1D under 25 °C and AM 1.5 solar spectrum and compared the experimental results. Poisson equation was used to describe the J-V characteristics and related to the electric field of p-n junction to the density of space charge as following (Eq. (1)):

$$\frac{\partial^2 \varphi}{\partial x^2} = -\frac{\partial E}{\partial x} = -\frac{\rho}{\epsilon_s} = -\frac{q}{\epsilon_s} [p - n + N_D(x) - N_A(x) \pm N_{def}(x)] \quad (1)$$

where  $\epsilon_s$  is permittivity of static materials,  $\varphi$  electrostatic potential, p is hole density, q: elementary charge and n: density of electrons. At steady state, the continuity equations of electron and hole as follows (Eqs. (2) and (3)):

$$\frac{\partial J_n}{\partial x} + G - U_n(n, p) = 0 \quad (2)$$

$$\frac{-\partial J_p}{\partial x} + G - U_p(n, p) = 0 \quad (3)$$

where  $J_n$  and  $J_p$  are the electrons and holes current density, G: generation rate of electron-hole and  $U_n$  and  $U_p$  are the rate of net recombination.

Equations of drift diffusion (Eqs. (4) and (5)):

$$J_n = q p \mu_n E + q D_n \frac{\partial n}{\partial x} \quad (4)$$

$$J_p = q p \mu_p E - q D_p \frac{\partial p}{\partial x} \quad (5)$$

$\mu_n$  and  $\mu_p$  are the mobility of electron and hole,  $D_n$ ,  $D_p$ : electron and hole diffusion coefficient and q: elementary charges. For the simulation of SCAPS, the parameters need to be entered. The solar cell parameters were taken depended on previously works [68,69] and summarized in Table 1.

#### 2.5. Characterization

The functional groups of prepared samples were investigated utilizing Fourier transform infrared spectra (PerkinElmer, Germany), while the Raman spectra were carried out at room temperature by Raman spectrometer (Tensor II, Bruker, Germany) with excitation light source of 522 nm. The X-ray photoelectron spectroscopy measurements were determined to estimate binding energy with ESCALAB 250XI system. Crystal structure and size were characterized with Cu K $\alpha$  radiation ( $\lambda = 1.5418 \text{ \AA}$ ) via X-ray diffractometer

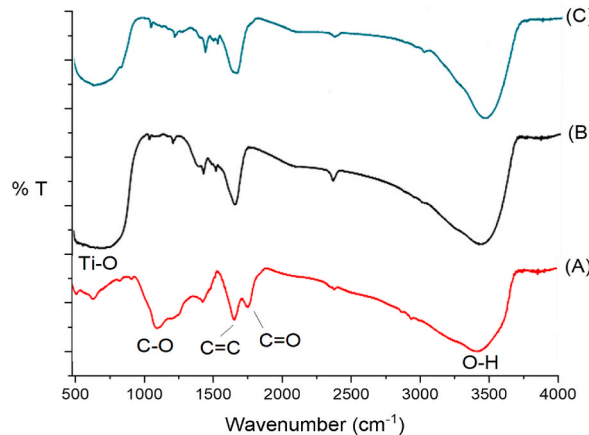


Fig. 3. FTIR of (A) GO, (B) TiO<sub>2</sub> and (C) 3 wt% GO-TiO<sub>2</sub>.

**Table 1**  
Perovskite solar cell parameters set.

Parameters	ITO	TiO <sub>2</sub>	GO/TiO <sub>2</sub>	perovskite	Spiro-oMeTAD
Thickness (nm)	290	30	280	290	220
Bandgap (eV)	3.4	3.12		1.54	3.06
Electron affinity (eV)	3.9	4		3.8	2.1
Dielectric perittivity (er)	8	9		6.2	2.8
Electron mobility (cm <sup>2</sup> /Vs)	18	18		1.8	10 <sup>-3</sup>
Hole mobility (cm <sup>2</sup> /Vs)	12	12		1.8	10 <sup>-3</sup>
Donor density (cm <sup>-3</sup> )	3*10 <sup>19</sup>	2*10 <sup>15</sup>		10 <sup>9</sup>	/
Acceptor density (cm <sup>-3</sup> )	/	/		10 <sup>9</sup>	3*10 <sup>19</sup>

(D8 advanced, Bruker, Germany) and using Scherrer equation [70] as following (Eq. (6)):

$$D = 0.9 \lambda / \beta \cos \theta \quad (6)$$

Morphology of the prepared samples was observed on a field emission electron microscope type (S-3400 N, Hitachi, Japan). The optoelectronic properties were estimated by using photoluminescence that recorded on fluorescence spectrophotometer (F-4600, Hitachi, Japan), while the time-resolved photoluminescence decay was recorded at 650 nm utilizing excitation light pulse with 470 nm and 5 MHz frequency from F-7000 spectrophotometer. The lifetime decay was calculated [71] by using following (Eq. (7)):

$$I(t) = I_0 + A_1 e^{\left(\frac{-t+\tau_1}{\tau_1}\right)} + A_2 e^{\left(\frac{-t+\tau_2}{\tau_2}\right)} \quad (7)$$

where  $A_1$ ,  $A_2$  are weight factors of decay channel,  $\tau_1$ ,  $\tau_2$  first and second order of decay time. The average lifetime was calculated by using the following equation (Eq. (8)):

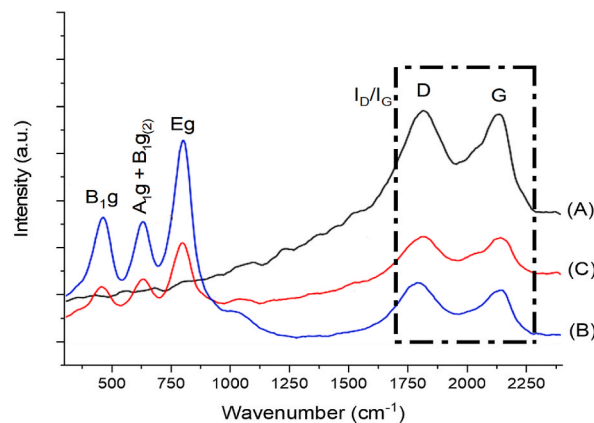
$$\tau_{avg} = \frac{\sum_n \sum_m A_n \tau_n^2}{\sum_n \sum_m A_n \tau_n} \quad (8)$$

### 3. Results and discussion

#### 3.1. Structure analysis

The structural investigations on the prepared compounds were carried out using FTIR, Raman spectra, XRD and XPS. To confirm fully oxidize of graphite to GO, FTIR analysis was carried out. The FT-IR spectrum of GO, TiO<sub>2</sub>, and GO incorporated TiO<sub>2</sub> is shown in Fig. 3a–c. In Fig. 3a, a broad band located at 3410 cm<sup>-1</sup> can be attributed to OH stretching vibration mode. The stretching mode of C=O is shown at 1755 cm<sup>-1</sup>, while the skeleton vibration of C=C is observed at 1652 cm<sup>-1</sup>. At 1395 cm<sup>-1</sup>, a peak is clearly appeared, which is corresponding to binding mode of C–H. Another peak is noted at 1048 cm<sup>-1</sup>, which assigned to vibration of C–O [72]. The FTIR spectrum of TiO<sub>2</sub> anatase phase is shown in Fig. 3b. The main vibration peak is located at 625 cm<sup>-1</sup>, which assigned to Ti–O bond. In spectrum of GO incorporated TiO<sub>2</sub> (Fig. 3c), the absorption peak is weekend, which indicate some interaction between TiO<sub>2</sub> and GO, as well as, reduce in adsorption peaks of O–H and C=O and become weaker, which is corresponding to some reaction.

The Raman spectra of GO as well as TiO<sub>2</sub> particles modified with 1 wt% or 3 wt% GO are shown in Fig. 4a–c. Raman spectra of all synthesized compounds are demonstrated two eminent intensity peaks centered at 1380 and 1615 cm<sup>-1</sup>, which assigned to D and G



**Fig. 4.** Raman spectrum (A) GO, (B) 1 wt% and (C) 3 wt% GO incorporated TiO<sub>2</sub>.

bands, respectively. As shown in Fig. 4b and c, four predominate intensity bands are located at 180, 425, 540 and 663  $\text{cm}^{-1}$  which are attributed to  $E_g$ ,  $B_{1g}$ ,  $A_{1g} + B_{1g}$  and  $E_g$  active modes of anatase phase. When compared the intensity ratio  $I_D/I_G$  for 1 wt% and 3 wt% GO incorporated  $\text{TiO}_2$ , it is found that it increased with increase of GO dopant, which indicated formation of more carbonic structures with  $\text{sp}^3$  hybridization. The D intensity peak is shifted to lower wavenumber with increasing  $\text{TiO}_2$ , which may be back to the stress by  $\text{TiO}_2$  nanoparticles growing on the graphene sheet surface, this confirms more reaction or interaction between GO and  $\text{TiO}_2$  [73,74, 75–79].

To estimate the elemental composition and chemical state, XPS analysis was carried out. The XPS analyses of GO and 1 wt% GO- $\text{TiO}_2$  are shown in Fig. 5a–d, in which fully element fine spectra of C 1s and O 1s can be distinctly specified. The fine spectra of O 1s in GO and 1 wt% GO- $\text{TiO}_2$  are also shown in Fig. 5b. The XPS analysis shows difference in peak position and shapes. For GO, the intensity peak at 533 eV is imputed to OH groups on the surface (Fig. 5b), while the intensity peak at 530 eV can be corresponding to Ti–O–C bond for 1 wt% GO- $\text{TiO}_2$  and this bonding refers to strong interaction between GO and  $\text{TiO}_2$  that been product from incorporating process [80]. The C 1s intensity peak of GO and 1 wt% GO- $\text{TiO}_2$  are demonstrated in Fig. 5c. The results indicated that the binding energy C 1s for 1 wt% GO- $\text{TiO}_2$  is lower than in GO. Moreover, it appears two peaks at 286 and 288 eV which assigned to C–C and C=C of pure GO, while intensity peak centered at 284 eV corresponds to C–O of 1 wt% GO- $\text{TiO}_2$ . Fig. 5d shows deconvolution curves of C 1s for pure GO and 1 wt% GO- $\text{TiO}_2$ . The peaks located at 286, 288 and 290 eV are respectively corresponding to C–C, C–O and O–C=O bonds of GO, while for same bonds in 1 wt% GO- $\text{TiO}_2$ , it showed at 284, 286 and 288 eV [81]. As illustrated obviously in XPS spectra, the intensity strength of oxygen functional group reduced clearly in 1 wt% GO- $\text{TiO}_2$  indicating reducing it after incorporation process.

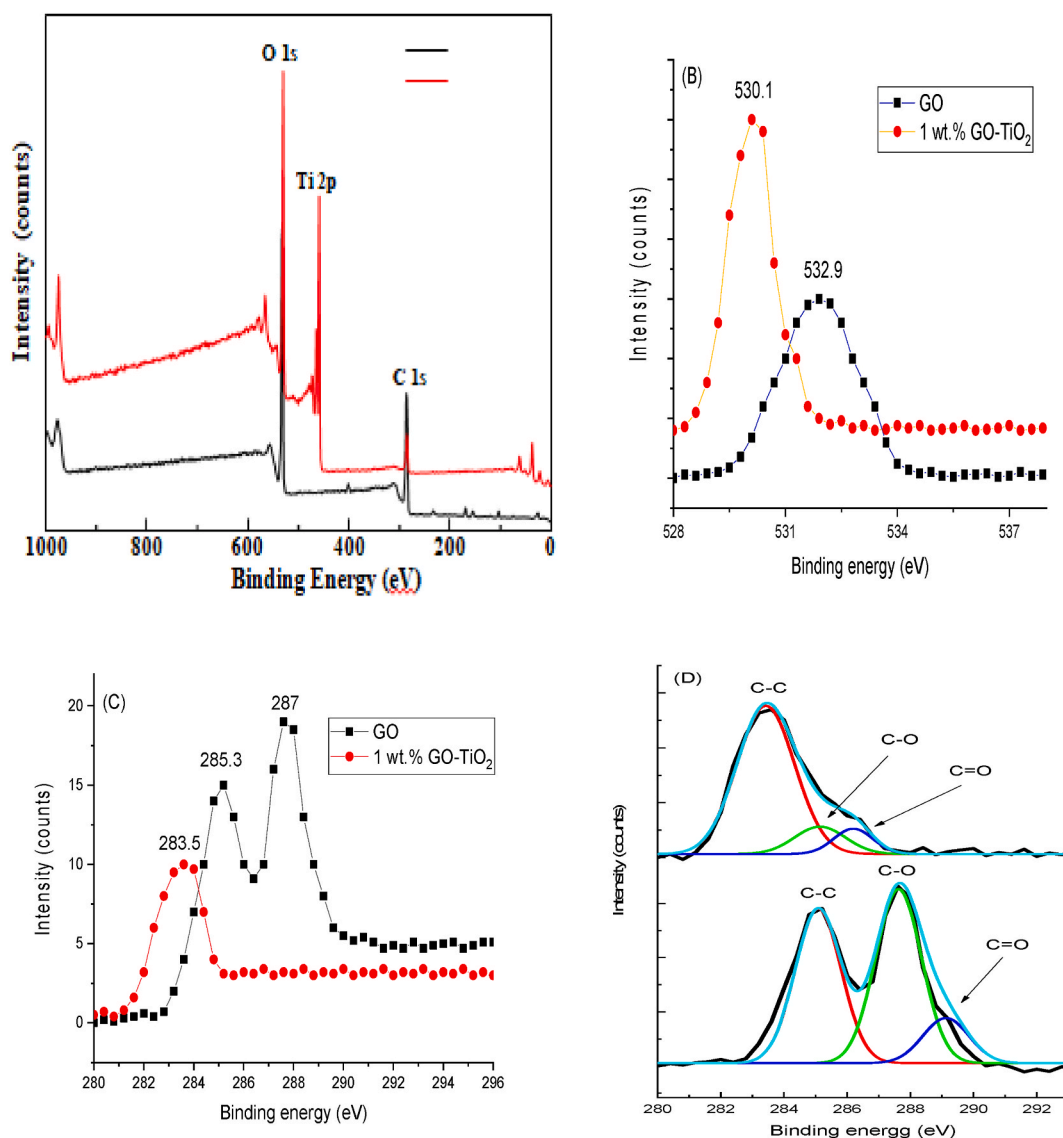


Fig. 5. XPS of GO and 1 wt% GO- $\text{TiO}_2$  (a) survey spectrum, (b) O 1s, (c) C 1s, and (d) deconvoluted C 1s.

The XRD of pure GO, TiO<sub>2</sub> and 1, 3 and 5 wt% GO incorporated TiO<sub>2</sub> nanocomposite are shown in Fig. 6a–e. The diffraction peak locates at 11.7° is assigned to GO, corresponding to the spacing of layer of graphene oxide (GO) 0.75 nm, which is larger than layer spacing of graphite that reported by Wang et al. [82]. The enlarged layer space is indicated entering oxygen-functional groups into the graphite structure, which leads to expansion of layer of graphite. The XRD pattern of TiO<sub>2</sub> and 1, 3 and 5 wt% GO incorporated TiO<sub>2</sub> are in good agreement with tetragonal anatase TiO<sub>2</sub>, the results with (JCPDS-21-1272), diffraction peaks at 2θ of 25.35°, 37.78°, 48.07°, 53.93°, 55.11° and 62.72° corresponding to (101), (004), (200), (105), (211) and (204), respectively. The XRD results suggest that incorporating GO holds the anatase TiO<sub>2</sub> phase profitably with no obvious effect of patterns of it, as well as, the results show no obvious diffraction peaks for GO in GO-TiO<sub>2</sub> nanocomposite, demonstrating the relatively limited mass percentage of GO incorporating [83]. Moreover, the expanded spectra obviously appeared at (101) diffraction peak at 2θ of 25.35°, which is progressively moving towards lower 2θ, indicating a possible reaction between GO and TiO<sub>2</sub> during incorporating process.

The XRD analysis of (FAPbI<sub>3</sub>)<sub>0.8</sub>(MAPbBr<sub>3</sub>)<sub>0.2</sub> thin film precipitated via different TiO<sub>2</sub> and 1 wt% GO-TiO<sub>2</sub> electron transfer materials is shown in Fig. 7a–c. The results appear diffraction peaks at 2θ of 14.3°, 21.12°, 29.55° and 33°, respectively corresponding to (111), (120), (222) and (231), which revealed the tetragonal structure perovskite lattice [84]. On the other hand, the results exhibited slight shift towards higher positions, with respect to 2θ of (14.3°) because of deposition of MAPbBr<sub>3</sub>. The XRD patterns appeared that the reflection peak at 14.3 (111) in GO-TiO<sub>2</sub> based perovskite film is higher than TiO<sub>2</sub> and it preferred (111) orientation axis. Moreover, for GO-TiO<sub>2</sub> based perovskite film, the relative intensity of planes (111) to (120) is much higher than other film, which may be back to the vertical growth perovskite grain.

### 3.2. Morphology characterization

The surface morphology of the prepared samples was estimated using FESEM-EDX and TEM. The FESEM images of GO and rGO are shown in Fig. 8a and b. The images are exposed thin sheets with folded morphology, as well as, the GO images shown a small rolling structure and the aggregated GO sheets are straggled from one another. After reduction process (Fig. 8b), the layer of rGO becomes less carbonic, illustrating the presence of well exfoliated graphene. Moreover, the rGO SEM images demonstrated a very small thickness and random distribution [85]. Fig. 8c appears the FESEM images of TiO<sub>2</sub> were precipitated at 2 ml/h feed rate, which disclosed forming NFs with uniform diameter and length. The FESEM images of incorporating different weight ratio of GO with TiO<sub>2</sub> are exhibited in Fig. 8d–f. The images appear that the GO ratio incorporating are effect on the diameter of TiO<sub>2</sub> NFs clearly, whereas, the diameter of TiO<sub>2</sub> NFs is reduced upon increase of GO, but decreased from 135 to 230 nm. Fig. 8g–j shows the FESEM images of (FAPbI<sub>3</sub>)<sub>0.8</sub>(MAPbBr<sub>3</sub>)<sub>0.2</sub> thin film precipitated on the FTO substrate, TiO<sub>2</sub>, 3 wt% rGO-TiO<sub>2</sub> and 5 wt% rGO-TiO<sub>2</sub> respectively. The images (Fig. 8g) exhibit a consolidated and bushy or dense with 450–500 nm grain size, while Fig. 8h shows severe in grain size up to 600 nm and these boosts in grain size propose that the TiO<sub>2</sub> eases nucleation site for the (FAPbI<sub>3</sub>)<sub>0.8</sub>(MAPbBr<sub>3</sub>)<sub>0.2</sub>. When (FAPbI<sub>3</sub>)<sub>0.8</sub>(MAPbBr<sub>3</sub>)<sub>0.2</sub> is precipitated on FTO/TiO<sub>2</sub>, the grain size has been rising to 800 nm approximately with consolidated in nature as shown in Fig. 8i. This increase in particle size and compact nature of grain size causes higher VOC, FF and lower recombination rate, leading to higher

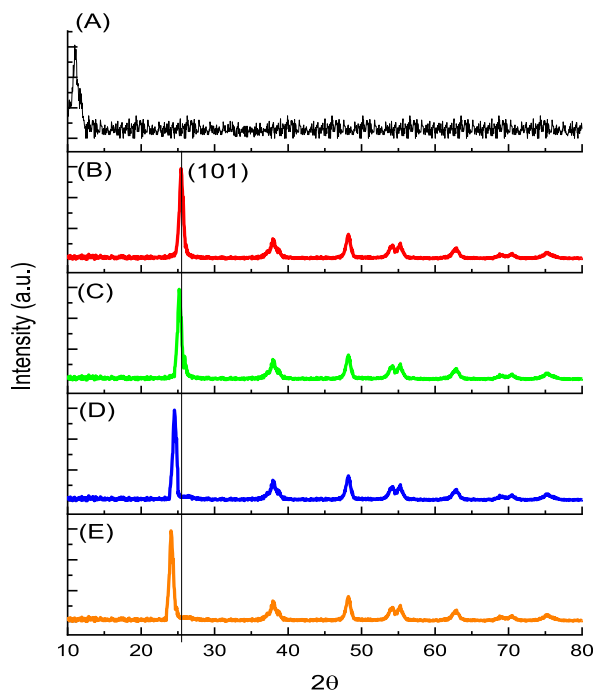


Fig. 6. XRD of (A) GO, (B) TiO<sub>2</sub>, (C) 1 wt% GO-TiO<sub>2</sub>, (D) 3 wt% GO-TiO<sub>2</sub> and (E) 5 wt% GO-TiO<sub>2</sub>.

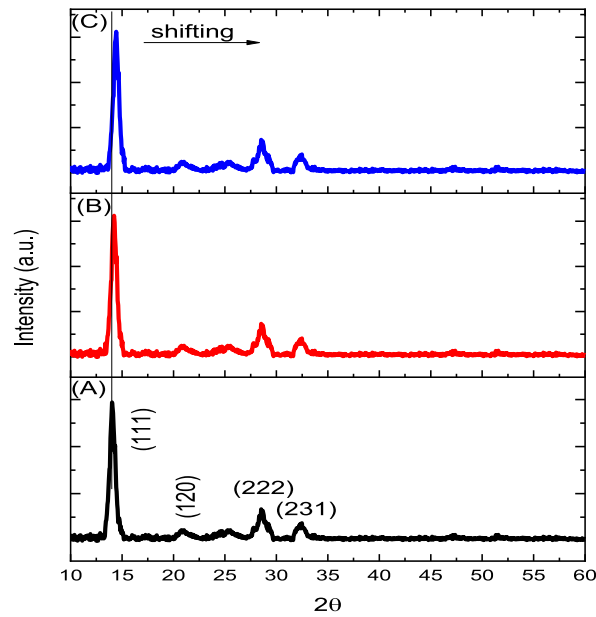


Fig. 7. XRD of (A)  $(\text{FAPbI}_3)_{0.8}(\text{MAPbBr}_3)_{0.2}$ , (B)  $\text{TiO}_2$  doped and (C) 1 wt%  $\text{GO-TiO}_2$  doped.

conversion efficiency of solar cell. Moreover, an increase in rGO incorporating ratio to 5 wt% causes slight reduction in grain size to 750 nm, but most of grain remain precise, as in Fig. 8j. These reduction in grain size emerges from high nuclei center that is produced from higher concentration of GO and smaller diameter of  $\text{TiO}_2$  [86]. From these FESEM results, the 3 wt% rGO- $\text{TiO}_2$  electron transfer materials eases the less sites of nucleation. It is recognized that lower crystallization of  $(\text{FAPbI}_3)_{0.8}(\text{MAPbBr}_3)_{0.2}$  perovskite layer consisted of a higher  $\Delta G_{\text{critical}}$  energy, which then decreases the nuclei numbers. Therefore, the growth of perovskite layer will begin from limited defined nuclei centers, which results in limited grain boundary and grain with large size [87,88].

### 3.3. Optoelectronic analysis

The photoluminescence (PL) and time-resolved photoluminescence (TRPL) analysis were carried out to estimate the optoelectronic properties and charge dynamic of  $\text{TiO}_2$ , 1 and 3 wt%  $\text{GO-TiO}_2$  deposited on  $(\text{FAPbI}_3)_{0.8}(\text{MAPbBr}_3)_{0.2}$  layer as an electron transfer material (ETMs) and the results are shown in Figs. 9 and 10. In Fig. 10, the PL spectra appeared without a hole transfer material (HTM) quencher, as well as, it showed an emission peak at  $\sim 778$  nm back to prepared samples. The PL spectra showed emission peak at  $\sim 778$  nm for  $\text{TiO}_2\text{NFs}/(\text{FAPbI}_3)_{0.8}(\text{MAPbBr}_3)_{0.2}$  with a reduction for hybrid samples because of the effective electron disarmament from conduction band (CB) of  $(\text{FAPbI}_3)_{0.8}(\text{MAPbBr}_3)_{0.2}$  to CB of GO and then to  $\text{TiO}_2$ . In addition to, the PL intensity of the produced nanoparticles dropped when the concentration of GO increased, showing a delay or friction in electron-hole recombination due to effective charges transfer as the doping ratio was increased.

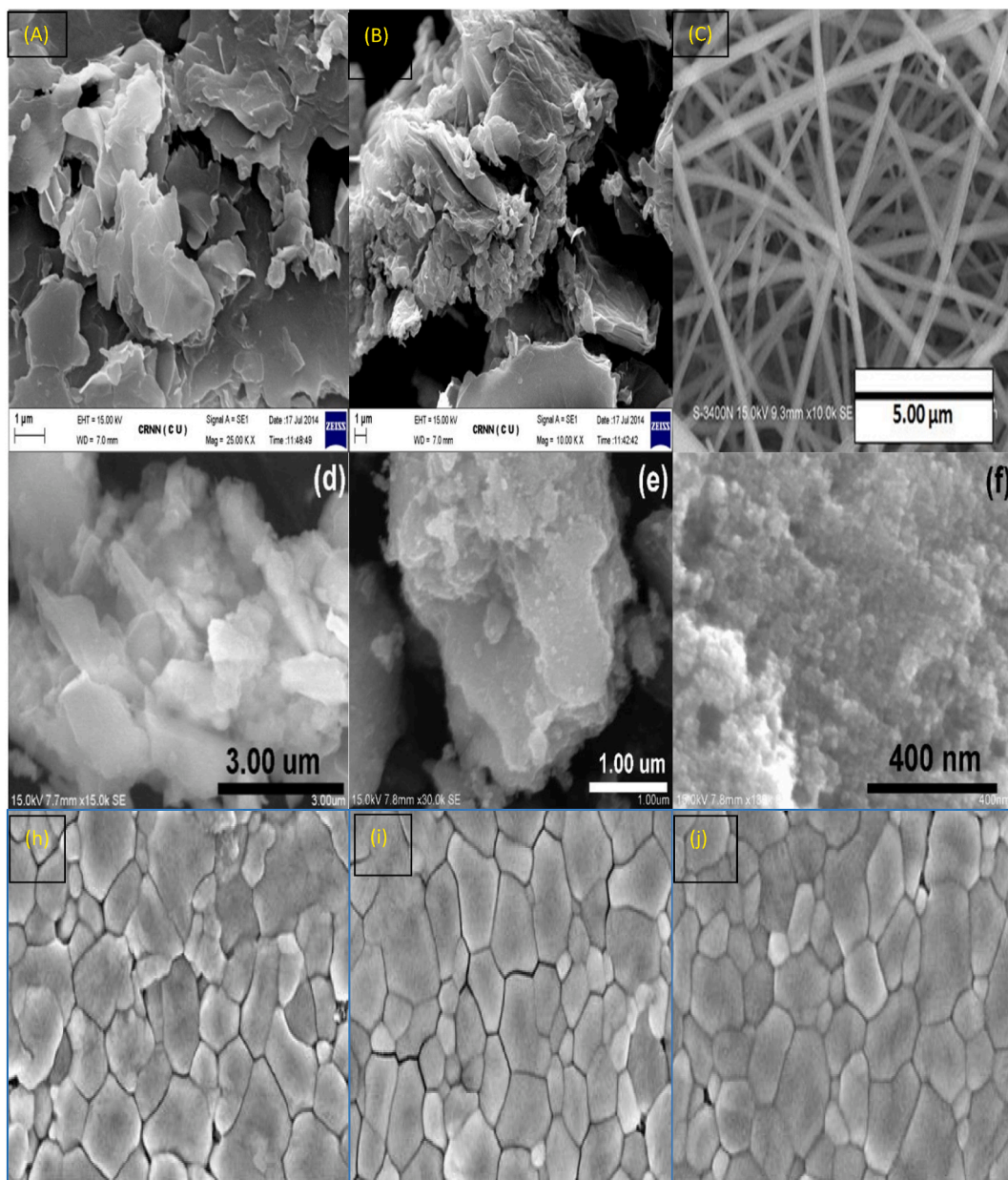
Time-resolved photoluminescence (TRPL) analysis was measured to estimate the electron lifetime of the prepared electron transfer materials (ETMs) and the results are shown in Fig. 10. The spectrum was fitted to the function of tri-exponential (Eq. (2)) and it showed the decay of lifetime in nanoseconds. This function contains a rapid decay lifetime component, which would generate from the charge carriers quenching that relocated to the electron contacts, while the two slow lifetime of decay component could be assign to the relative recombination of holes and electrons before collection of charges [88]. The average lifetimes of the prepared samples were calculated from Eq. (3) and the results are summarized in Table 2.

The results indicate that the  $\text{TiO}_2$  and 3 wt%  $\text{GO-TiO}_2$  deposited on  $(\text{FAPbI}_3)_{0.8}(\text{MAPbBr}_3)_{0.2}$ -layer film electrodes demonstrate fast decay lifetimes of 290 and 86 ns ( $\tau_1$ ) respectively, while slow lifetimes of decay for respective electrodes are 13 and 19 ns ( $\tau_2$ ). The estimated fraction values  $A_1$  of the fast decay of lifetimes are 0.037 and 0.026 for  $\text{TiO}_2/(\text{FAPbI}_3)_{0.8}(\text{MAPbBr}_3)_{0.2}$  and 3 wt%  $\text{GO-TiO}_2/(\text{FAPbI}_3)_{0.8}(\text{MAPbBr}_3)_{0.2}$  electrodes and 0.732 and 0.626 for  $A_2$  of respective electrodes. These results suggest that electrons extractions from conduction bands of  $(\text{FAPbI}_3)_{0.8}(\text{MAPbBr}_3)_{0.2}$  to CB of 3 wt%  $\text{GO-TiO}_2$  is more effective relatively, as well as, it exposed that GO eases effective charge dissociation in the  $(\text{FAPbI}_3)_{0.8}(\text{MAPbBr}_3)_{0.2}$  layer. In addition to, the TRPL measurements illustrated that the incorporating nanocomposite sample eases electron extraction capability from CB of  $(\text{FAPbI}_3)_{0.8}(\text{MAPbBr}_3)_{0.2}$  layer to CB of 3 wt%  $\text{GO-TiO}_2$ .

### 3.4. Numerical investigation of solar cell

Beginning from the enter parameters of fabricated solar cell layers, the J-V properties of fabricated PVSC were emulated and paralleled to experimental data. A theoretical treatise of absorber defect density effect on the properties of PVSC is first carried out. The





**Fig. 8.** FESEM of (a) GO, (b) rGO, (c) TiO<sub>2</sub> NFs, (d) 1 wt% GO-TiO<sub>2</sub>, (e) 3 wt% GO-TiO<sub>2</sub>, (f) 5 wt% GO-TiO<sub>2</sub>, (FAPbI<sub>3</sub>)<sub>0.8</sub>(MAPbBr<sub>3</sub>)<sub>0.2</sub> perovskite deposited on different ETLs on (h) 1 wt% GO-TiO<sub>2</sub>, (i) 3 wt% GO-TiO<sub>2</sub> and (j) 5 wt% GO-TiO<sub>2</sub>.

density defect is a substantial parameter that defines the electrical properties of PVSCs and, hence can assist to detect the efficiency [89]. In the range of  $10^{14}$ - $10^{17}$ , the effect results of defect densities on PVSC J-V properties are displayed in Fig. 11a. To detect the defect density that reproduce the experimental data, emulsion date or results were contrasted with experimental results. The defects density has a considerable effect on the value of VOC, were it reduce from 1.15V to 0.85V upon increase in defect density from  $10^{14}$  to  $10^{17}$   $\text{cm}^{-1}$  without any impact of Jsc values, and these results are in agreement with reported results [90]. The experimental and theoretical calculation of J-V plot are presented in Fig. 11b. The results show that the calculations are in agreement with each other. The experimental and theoretical calculations appeared that conversion efficiency of power were 13.15% and 13.32% respectively. on the other hand, the quantum efficiency is another substantial property to investigate our numerical calculations, and the results are shown in Fig. 11c. The results obtained that the theoretical plot is in good agreement with experimental curve. As well as, from this plot, observed the contribution of electron-hole pairs that photogenerated in current density, where the experimental data appear 49% and 52% for theoretical data, which indicate that the half of photons fallen on PVSC are utilized in the process of conversion. From Fig. 11b and c, its illustrated that the design model depend on recombination during defects is fit to demonstrate experimental data

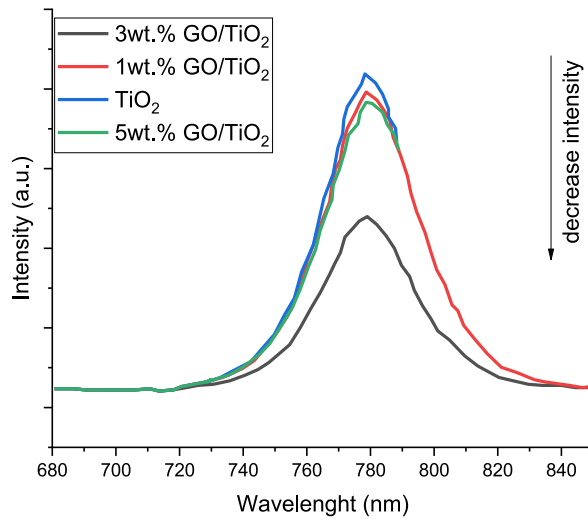


Fig. 9. PL spectrum of prepared electrodes.

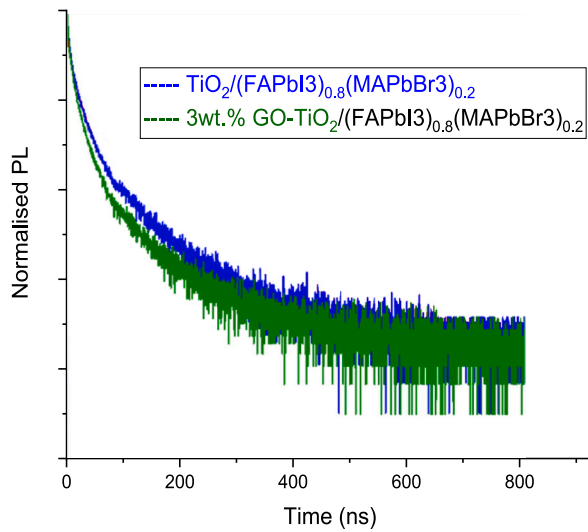


Fig. 10. TRPL of  $\text{TiO}_2/(\text{FAPbI}_3)_{0.8}(\text{MAPbBr}_3)_{0.2}$  and 3 wt%  $\text{GO-TiO}_2/(\text{FAPbI}_3)_{0.8}(\text{MAPbBr}_3)_{0.2}$ .

Table 2

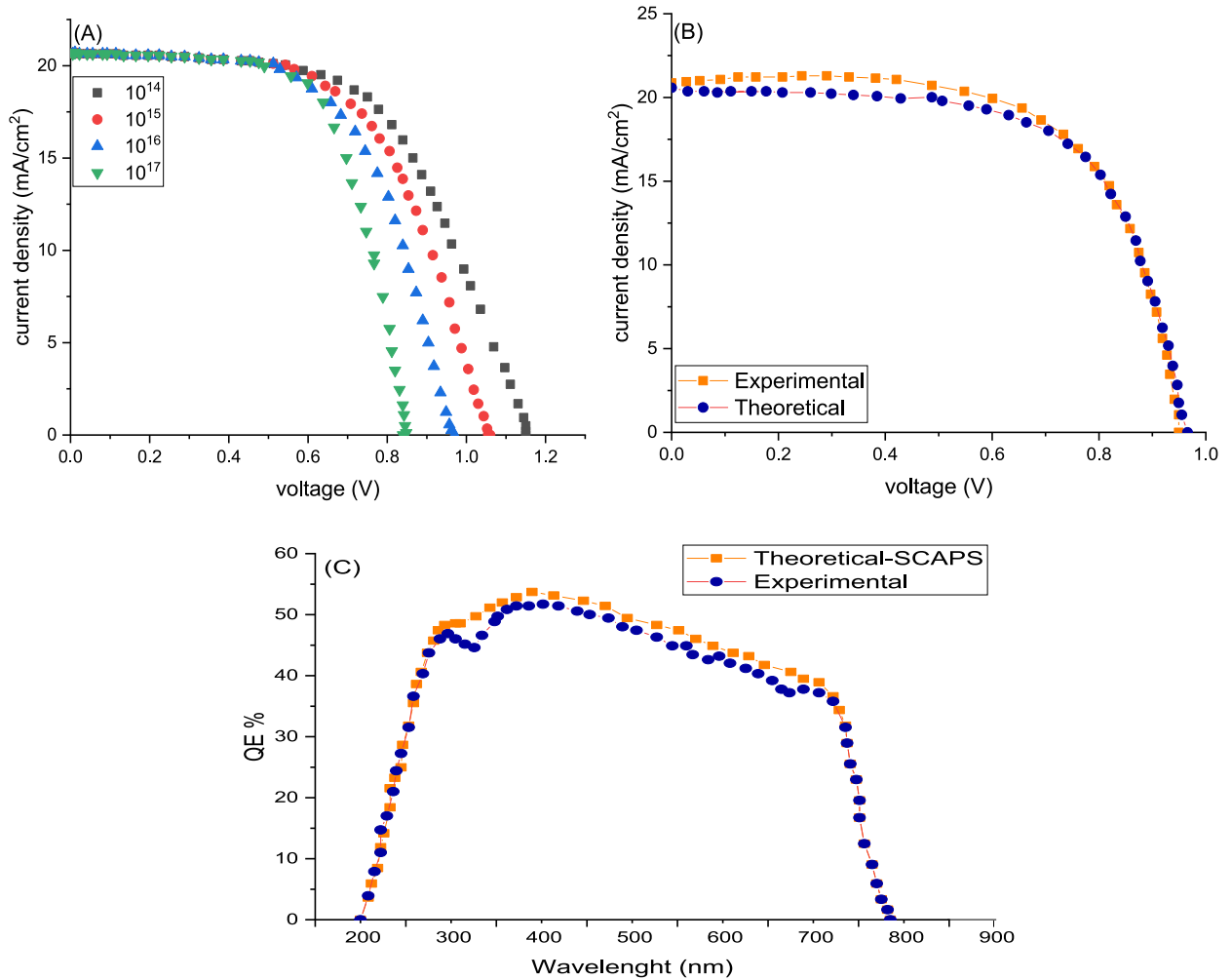
TRPL decay measurements of  $\text{TiO}_2$  and 3 wt%  $\text{GO-TiO}_2/(\text{FAPbI}_3)_{0.8}(\text{MAPbBr}_3)_{0.2}$ .

Perovskite/ETMs	$\tau_1$ (ns)	$\tau_2$ (ns)	$A_1^*$	$A_2^*$	$\langle \tau \rangle$ (ns)
$\text{TiO}_2/(\text{FAPbI}_3)_{0.8}(\text{MAPbBr}_3)_{0.2}$	290	13	0.037	0.732	151
3 wt% $\text{GO-TiO}_2\text{NFs}/(\text{FAPbI}_3)_{0.8}(\text{MAPbBr}_3)_{0.2}$	86	19	0.026	0.626	52

from QE and J-V data.

### 3.5. Photovoltaic measurements

Fig. 12a appear a graphical display of the perovskite system having Glass/ITO/3 wt% $\text{GO-TiO}_2/(\text{FAPbI}_3)_{0.8}(\text{MAPbBr}_3)_{0.2}/\text{MeO-TAD}/\text{Ag}$  arrangement utilized in this study. In fabricating PVSCs,  $(\text{FAPbI}_3)_{0.8}(\text{MAPbBr}_3)_{0.2}$  absorber layer is sandwiched between  $\text{TiO}_2$  block layer/3 wt% $\text{GO-TiO}_2$  that work as ETMs and spiro-MeOTAD HTMs. The SEM cross section image of fabricated PVSC is shown in Fig. 12b, where the thickness are 50, 130, 550, 200 and 70 nm for  $\text{TiO}_2$ , 3 wt% $\text{GO-TiO}_2$ ,  $(\text{FAPbI}_3)_{0.8}(\text{MAPbBr}_3)_{0.2}$  capping layer, HTM



**Fig. 11.** (a) effect of defect density on J-V curves of fabricated PVSC, (b) J-V plots from theoretical validation and experimental on PVSC, (c) QE of theoretical and experimental on PVSC.

and Ag respectively. The image is shown the production of  $(\text{FAPbI}_3)_{0.8}(\text{MAPbBr}_3)_{0.2}$  capping layer with highly crystalline and pin-hole free on the 3 wt%GO-TiO<sub>2</sub> film and this layer separates of the spiro and TiO<sub>2</sub> electron transfer materials (HTMs). So, in this device, the ohmic contact between electrodes are little possibilities. The current-voltage (J-V) properties of prepared PVCS and its parameters are shown in Fig. 12c and summarized in Table 3. In this report, we utilized four synthesized materials (TiO<sub>2</sub>, 1 wt% GO-TiO<sub>2</sub>, 3 wt% GO-TiO<sub>2</sub> and 5 wt% GO-TiO<sub>2</sub>) as electron transporting materials (ETMs) and investigated its execution utilizing the  $(\text{FAPbI}_3)_{0.8}(\text{MAPbBr}_3)_{0.2}$  as adsorbing layer of light. The results appeared that the device based on TiO<sub>2</sub> showed the efficiency less than device-based GO-TiO<sub>2</sub>, which may be attributed to smaller grain size as showed in FESEM results [91]. From results (Table 3), the device based 1 wt% GO-TiO<sub>2</sub> shown VOC of 1.124 V, J<sub>SC</sub> of 23.45 mAcm<sup>-2</sup>, FF of 0.702 yielding PCE 18.50%, while in device based 3 wt% GO-TiO<sub>2</sub> exhibited VOC of 1.76 V, J<sub>SC</sub> of 24.06 mAcm<sup>-2</sup>, FF of 0.712 yielding PCE 20.14%. The difference in PCE may be back to the fast transportation of electrons and low recombination of e<sup>-</sup>-h<sup>+</sup> [91]. In the case of 5 wt% GO-TiO<sub>2</sub> ETMs, the J-V results shown VOC of 1.133 V, J<sub>SC</sub> of 23.59 mAcm<sup>-2</sup>, FF of 0.734 yielding PCE 19.61%. The decrease in efficiency emerges because of the high graphene oxide concentration and low grain size, and this causes decreasing in crystalline and hence difficultly in charge transport. All efficiency results are in agreement with PL spectrum results.

To distinctly appear the pathway of charge transfer, the band gap diagram of ITO, TiO<sub>2</sub>, GO-TiO<sub>2</sub>, ETM and ETM was plotted on the bases of their edges of conduction band and energy gap as following:

Anatase TiO<sub>2</sub> (-4.0eV), ETM layer (-3.9eV), the HOMO level of ETM layer (-5.20eV) and (-4.3eV) for GO as shown in Fig. 13. The diagram shown the reduce or lower energy gap of GO and this causes transfer the photoexcited electron from ETM layer and TiO<sub>2</sub> to GO. So, the GO/TiO<sub>2</sub> nanocomposite based PVSC should supply improved collection and charge transfer properties, as well as, a good performing wt.% GO/TiO<sub>2</sub> nanocomposite PVSC is fabricated via future controlling film thickness.

Comparing with previous reports (Table 4), this fabricated PVSCs showed a good J<sub>sc</sub>, VOC, and efficiency.

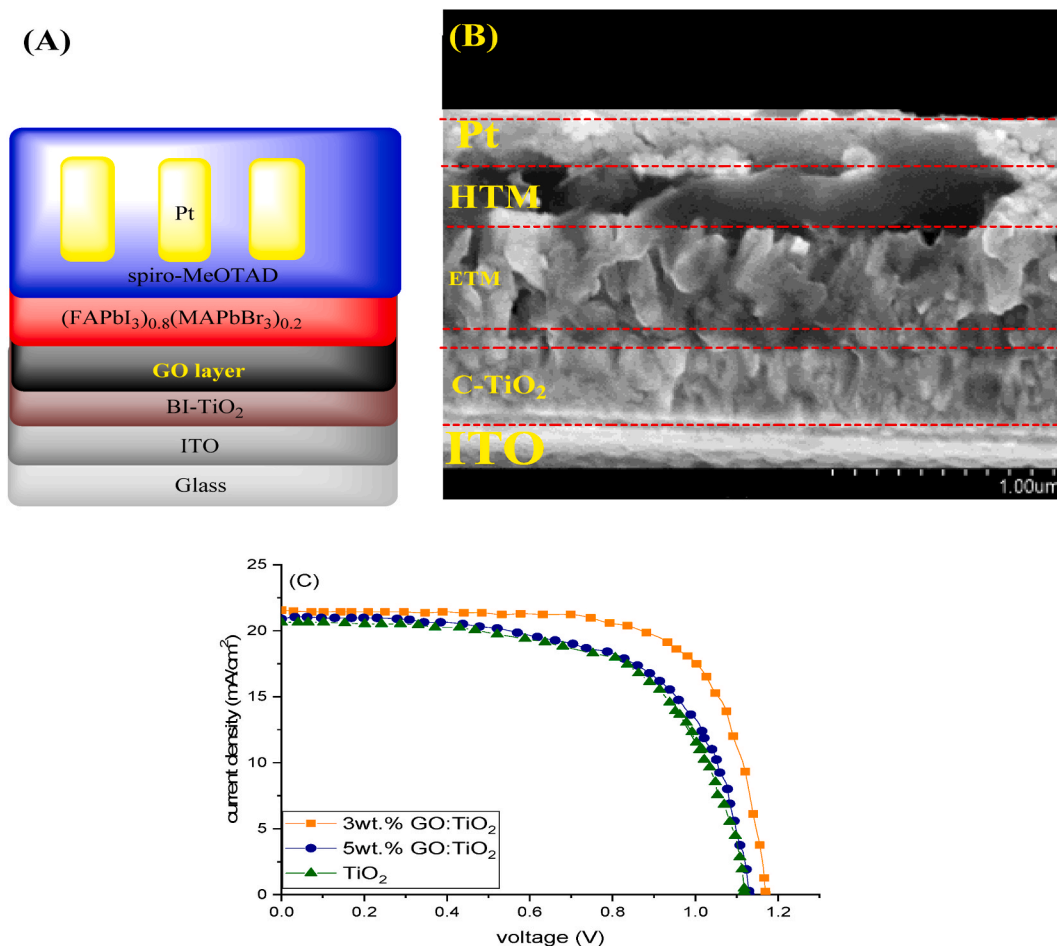


Fig. 12. (a) schematics device, (b) FESEM cross section of 3 wt% GO-TiO<sub>2</sub> based (FAPbI<sub>3</sub>)<sub>0.8</sub>(MAPbBr<sub>3</sub>)<sub>0.2</sub> and (c) J-S curve of fabricated PVSCs.

Table 3

J-V of prepared materials based on (FAPbI<sub>3</sub>)<sub>0.8</sub>(MAPbBr<sub>3</sub>)<sub>0.2</sub>.

Perovskite/ETMs	VOC (V)	Jsc (mA/cm <sup>2</sup> )	FF	%PEC
TiO <sub>2</sub> /(FAPbI <sub>3</sub> ) <sub>0.8</sub> (MAPbBr <sub>3</sub> ) <sub>0.2</sub>	1.132	23.73	0.686	18.42
1 wt% GO-TiO <sub>2</sub> NFs/(FAPbI <sub>3</sub> ) <sub>0.8</sub> (MAPbBr <sub>3</sub> ) <sub>0.2</sub>	1.124	23.45	0.702	18.50
3 wt% GO-TiO <sub>2</sub> NFs/(FAPbI <sub>3</sub> ) <sub>0.8</sub> (MAPbBr <sub>3</sub> ) <sub>0.2</sub>	1.176	24.06	0.712	20.14
5 wt% GO-TiO <sub>2</sub> NFs/(FAPbI <sub>3</sub> ) <sub>0.8</sub> (MAPbBr <sub>3</sub> ) <sub>0.2</sub>	1.133	23.59	0.734	19.61

#### 4. Conclusion

A series of wt.% GO-TiO<sub>2</sub> composite NFs high crystalline are synthesized by a electrospun method and deposited on boundary of (FAPbI<sub>3</sub>)<sub>0.8</sub>(MAPbBr<sub>3</sub>)<sub>0.2</sub> perovskite and its photovoltaic activity is investigated. The effect of GO in TiO<sub>2</sub> NFs have been estimated in detail. The 3 wt% GO-TiO<sub>2</sub> composite NFs eases the limited centers of nuclei cause in highly uniform perovskite grain size. The implied effect of the 3 wt% GO-TiO<sub>2</sub> and mechanism were studied by TRPL analysis. The results of charges transfer exposed that GO eases better collection of charges and fast electron transportation, causes to a frequently ameliorated PEC. The optimized 3 wt% GO-TiO<sub>2</sub> composite NFs-based electron transfer materials obtained 20.14 % PCE with VOC 1.176 V, Jsc 24.06 mA/cm<sup>2</sup> and FF 0.712. The 3 wt% GO-TiO<sub>2</sub> could avail as an effective channel of electrons extraction and transfer from (FAPbI<sub>3</sub>)<sub>0.8</sub>(MAPbBr<sub>3</sub>)<sub>0.2</sub> perovskite layer. The hybridization also improved the values of Jsc and VOC, leading to higher efficiency transportation, as constructed to the TiO<sub>2</sub> nanoparticles. This novel synthesis method proposes an appropriate and friendly model to enhance the PVSC efficiency.

#### Data availability

All data generated or analyzed during this study are included in this published article.

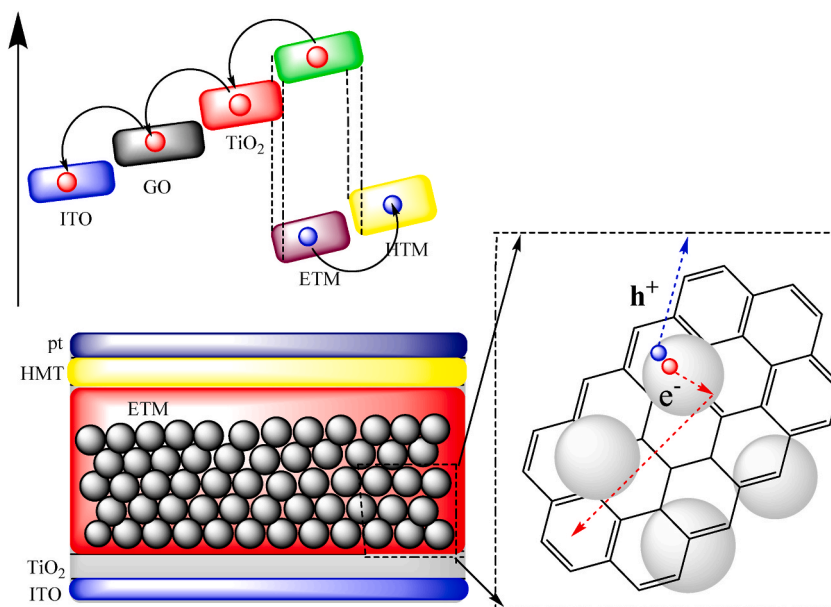


Fig. 13. Electron-hole transfer in fabricated PVSCs.

Table 4

Efficiency of PVSCs based on literature and comparison with this study.

Solar cell	Jsc (mA/cm <sup>2</sup> )	VOC (V)	Efficiency (%)	References
1wt. GO-TiO <sub>2</sub>	22.84	/	13.60%	[28]
GO-TiO <sub>2</sub>	22.78	0.61	17.73%	[35]
0.4 vol% rGO/TiO <sub>2</sub>	21.00	0.91	13.5	[36]
GO/TiO <sub>2</sub>	1	1	2.50	[37]
CH <sub>3</sub> NH <sub>3</sub> PbI <sub>3</sub> /TiO <sub>2</sub>	20.0	0.98	12.00	[38]
CH <sub>3</sub> NH <sub>3</sub> PbI <sub>2</sub> Br/TiO <sub>2</sub>	10.12	0.82	4.80	[39]
TiO <sub>2</sub> /(FAPbI <sub>3</sub> )0.8(MAPbBr <sub>3</sub> )0.2	1.13	23.73	18.42	Our study
3 wt% GO- TiO <sub>2</sub> NFs/(FAPbI <sub>3</sub> )0.8(MAPbBr <sub>3</sub> )0.2	1.17	24.06	20.14	Our study

#### CRedit authorship contribution statement

**Chou-Yi Hsu:** Writing – review & editing, Writing – original draft, Investigation, Data curation. **H.N.K. AL-Salman:** Writing – review & editing, Writing – original draft, Investigation, Data curation. **Hussein H. Hussein:** Hussein, Writing – review & editing, Writing – original draft, Investigation, Data curation. **Nizomiddin Juraev:** Writing – review & editing. **Zaid H. Mahmoud:** Writing – review & editing, Writing – original draft, Investigation, Data curation. **Saeb Jasim Al-Shuwaili:** Writing – review & editing. **Hanan Hassan Ahmed:** Writing – review & editing, Writing – original draft, Investigation, Data curation. **Ahmed Ali Ami:** Writing – review & editing. **Nahed Mahmood Ahmed:** Writing – review & editing, Writing – original draft, Investigation, Data curation. **Seitkhan Azat:** Writing – review & editing. **Ehsan kianfar:** Writing – review & editing, Writing – original draft, Investigation, Data curation.

#### Declaration of competing interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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